



# Computational investigation of chalcogenide spinel conductors for all-solid-state Mg batteries†

Julius Koettgen,<sup>a</sup> Christopher J. Bartel<sup>b</sup> and Gerbrand Ceder<sup>\*ab</sup>

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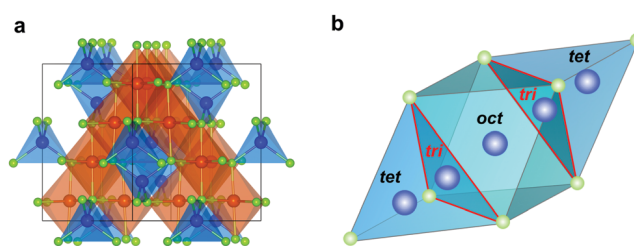
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**Seven  $\text{MgLn}_2\text{X}_4$  (Ln = lanthanoid, X = S, Se) spinels are calculated with density functional theory to have low barriers for Mg migration (< 380 meV) and are stable or nearly stable (within 50 meV per atom of stability with respect to competing structures and compositions). As the size of the Ln increases, Mg mobility is found to increase, but stability in the spinel structure is found to decrease.**

Magnesium batteries are an interesting alternative to Li-ion technology because of the potential for improved safety and higher energy density.<sup>1–3</sup> The ability to use Mg metal as an anode is beneficial because of its high volumetric energy density (3830 mA h mL<sup>−1</sup>) relative to that of graphite (700 mA h mL<sup>−1</sup>), which is generally employed as an anode in Li-ion batteries using liquid or polymer gel electrolytes.<sup>4</sup> Only a few good liquid electrolytes exist for Mg batteries as Mg metal cannot be deposited reversibly in electrolytes that react with the metal. For this reason, we recently explored whether solid electrolytes with sufficient Mg-ion conductivity exist.<sup>5</sup> Unlike for Li and Na, for which a large number of solids with very high ionic conductivity have been reported,<sup>6,7</sup> virtually no solids with high Mg-ion conductivity have been identified, apart from a few so-far unconfirmed computational predictions.<sup>8</sup>

Materials with high Li conductivity have been shown to possess low Mg mobility because the higher charge density of divalent Mg relative to that of monovalent Li leads to stronger electrostatic interactions with the host lattice.<sup>9</sup> Recently, two design principles to improve mobility in multivalent solids have been proposed. First, structures where the multivalent ion sits in a site where it has an unfavourable coordination environment have been observed to have lower migration barriers, in particular when the activated state is closer to the



**Fig. 1** (a)  $\text{MgLn}_2\text{X}_4$  spinel structure where Mg is blue, the lanthanoid (Ln) is orange, and the chalcogenide (X) is light green, (b) Diffusion path for  $\text{Mg}^{2+}$  (blue) from the tetrahedral site, tet, through the face-sharing triangular state, tri, then the octahedral site, oct. The diffusion path is locally symmetric such that before returning to tet,  $\text{Mg}^{2+}$  again passes through tri.

preferred coordination.<sup>7,9–11</sup> Superior  $\text{Mg}^{2+}$  mobility was previously observed in spinels (Fig. 1a) with composition  $\text{TM}_2\text{X}_4$  (TM = Ti, Cr, Mn, Fe, Co, Ni and X = O, S), where the diffusing ion resides in a tetrahedral site but migrates through an activated octahedral coordination as shown in Fig. 1b.<sup>9,10,12–14</sup> Second,  $\text{Mg}^{2+}$  mobility is generally higher for sulphides than oxides, suggesting that the inclusion of larger, more polarizable anions helps decrease the electrostatic interaction between the working ion and the host structure.<sup>10,14</sup>

These design principles led to the investigation of magnesium spinel chalcogenides  $\text{MgM}_2\text{X}_4$  with M = Sc, In, Y and X = S, Se, Te.<sup>5</sup> Several of these compositions have been reported experimentally, including  $\text{MgSc}_2\text{S}_4$ ,<sup>15</sup>  $\text{MgIn}_2\text{S}_4$ ,<sup>15,16</sup>  $\text{MgSc}_2\text{Se}_4$ ,<sup>17</sup> and  $\text{MgY}_2\text{Se}_4$ .<sup>17</sup> The calculated barriers for  $\text{Mg}^{2+}$  migration generally decrease from sulphides to selenides to tellurides as both the unit cell volume and the polarizability of the anion increase. In  $\text{MgSc}_2\text{Se}_4$ , *ab initio* calculations and nuclear magnetic resonance experiments indicate a  $\text{Mg}^{2+}$  migration barrier of 380 meV, comparable to barriers in Li-ion cathode materials.<sup>18</sup> In this work, we demonstrate that  $\text{Mg}^{2+}$  mobility can be further increased through the incorporation of large cations from the lanthanide series. However, destabilization of the spinel structure occurs upon increasing the size of the lanthanoid; therefore, the optimal spinel Mg conductor balances high mobility and synthesizability.

<sup>a</sup> Materials Sciences Division, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, USA. E-mail: gceder@berkeley.edu

<sup>b</sup> Department of Materials Science and Engineering, University of California, Berkeley, CA 94720, USA

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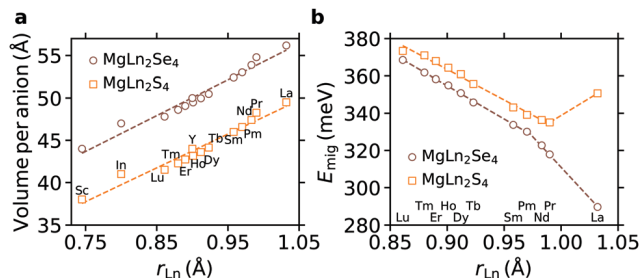


Fig. 2 (a) Effect of Ln ionic radius,  $r_{Ln}$ , on the calculated unit cell volume of the spinel structure, (b) Effect of  $r_{Ln}$  on the vacancy-mediated  $Mg^{2+}$  migration barrier,  $E_{mig}$ . Ionic radii are from Shannon.<sup>22</sup> Results for Sc, In, and Y are from ref. 5. In both panels, the dashed lines are drawn to guide the eye.

The  $Mg^{2+}$  mobility and stability of magnesium chalcogenide spinels,  $MgLn_2X_4$ , with  $Ln = Lu, Tm, Er, Ho, Dy, Tb, Sm, Pm, Nd, Pr, La$  and  $X = S, Se$  were investigated using density functional theory (DFT) in the generalized gradient approximation (GGA)<sup>19</sup> and the nudged elastic band method (NEB)<sup>20</sup> for  $Mg^{2+}$  mobility calculations. Meta-GGA<sup>21</sup> was used for stability calculations. Substituting larger lanthanoids into the spinel increases the unit cell volume, as shown in Fig. 2a. A 40% increase of the cation radius,  $r_{Ln}$ , from 0.745 Å ( $Sc^{3+}$ )<sup>22</sup> to 1.032 Å ( $La^{3+}$ ),<sup>22</sup> leads to a unit cell volume increase of ~30%. Fig. 2b shows the calculated  $Mg^{2+}$  migration barrier in these spinels assuming vacancy-mediated diffusion. The increase in volume to accommodate larger cations does indeed correlate with an improved  $Mg^{2+}$  mobility, as the calculated migration barriers decrease from 375 meV for  $MgSc_2Se_4$ <sup>5</sup> to 290 meV for  $MgLa_2Se_4$ . In fact, the barriers calculated for all materials are lower than that of the previously synthesized  $MgSc_2Se_4$ . The  $Mg^{2+}$  migration barrier decreases nearly linearly with  $r_{Ln}$ , with the notable exception of  $MgLa_2S_4$ , which is associated with the destabilization of Mg in the octahedral state as discussed later.

The energy profiles along the  $Mg^{2+}$  migration path are shown in Fig. 3. All compounds show a global minimum with Mg in a tetrahedral site (tet), confirming that it is a locally stable site in these spinel structures. All compounds except  $MgLa_2S_4$  show a local minimum for the octahedral state (oct), and the

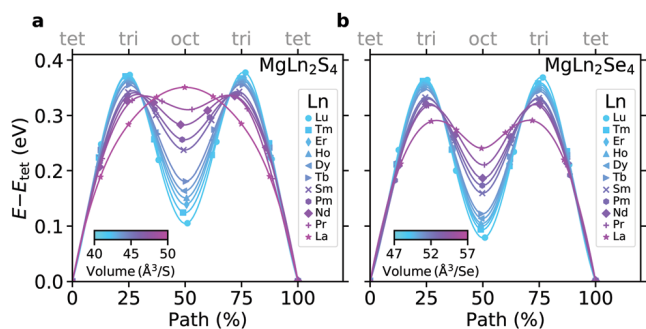


Fig. 3 Energy profiles for vacancy-mediated  $Mg^{2+}$  migration in sulphides (a) and selenides (b). Energy is shown relative to the tetrahedral Mg state ( $E - E_{tet}$ ). The colourbar provides the volume per anion of the pristine structure.

transition state (highest energy along the path) occurs when  $Mg^{2+}$  passes through the triangular face (tri) that is shared between tet and oct. Of particular importance for  $Mg^{2+}$  migration is the distance between the migrating  $Mg^{2+}$  and neighbouring chalcogenides (X), as the electron density of these anions impedes cation transport. Assuming regular polyhedra, the minimum distance between the centroid ( $Mg^{2+}$ ) and vertex ( $X^{2-}$ ) of each environment is proportional to  $\frac{d_{XX}}{\sqrt{3}} \approx 0.57d_{XX}$ ,  $\frac{3d_{XX}}{2\sqrt{6}} \approx 0.61d_{XX}$ ,

and  $\frac{d_{XX}}{\sqrt{2}} \approx 0.71d_{XX}$  for tri, tet, and oct, respectively, where  $d_{XX}$  is the X-X bond distance that forms the shared edge of the octahedron and tetrahedron. This relationship dictates that tri will have the smallest Mg-X distance and therefore likely be the highest energy (transition) state as long as oct and tet are local minima in the energy profile. For  $MgLa_2S_4$ , oct is not a local minimum and the migration barrier for this material is therefore determined by  $E_{oct} - E_{tet}$ .

The  $Mg^{2+}$  migration energy,  $E_{mig}$ , is compared to the energy of the octahedral state relative to the tetrahedral state ( $E_{oct} - E_{tet}$ ) in Fig. 4. The energies of the tri and oct states (relative to tet) change in opposite directions as the structure expands to accommodate larger and larger Ln's. tri is stabilized with increasing  $r_{Ln}$  because this state requires unfavourably small Mg-X distances as  $Mg^{2+}$  passes through the triangular face, and structure expansion lengthens this distance. The destabilization of oct when  $r_{Ln}$  is increased can be attributed to the relatively small size of  $Mg^{2+}$ . This is reflected in the ground-state structures for MgX compounds – MgS is rock salt with octahedral Mg coordination but increasing  $r_X$  to MgSe changes the preferred coordination to tetrahedral in wurtzite (ground-state structures determined with GGA). The preference for coordination of a cation, c, by an anion, a, can be estimated by classic radii ratio rules<sup>23</sup> where tetrahedral coordination is expected for  $0.225 < r_c/r_a \leq 0.414$  and octahedral coordination for  $0.414 < r_c/r_a \leq 0.732$ . For  $Mg^{2+}$ , the radius ratio is 0.391 for sulphides and 0.364 for selenides, suggesting  $Mg^{2+}$  should prefer large tetrahedral or small octahedral environments for these chalcogenides. The octahedral Mg-X bond distance in MgS is ~2.61 Å and expands from ~2.73 (2.85) Å in  $MgLu_2S_4$  ( $MgLu_2Se_4$ ) to ~2.85 (2.96) Å in

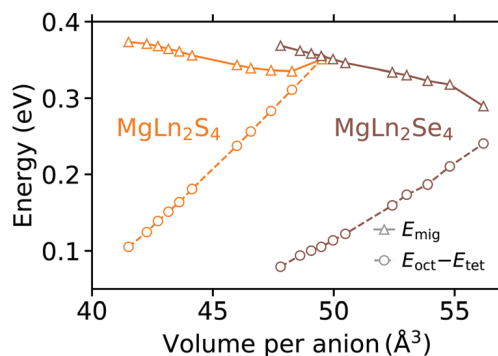


Fig. 4 Comparing the  $Mg^{2+}$  migration energy,  $E_{mig}$ , to the difference in octahedral and tetrahedral energies,  $E_{oct} - E_{tet}$  as a function of the calculated volume per anion in the pristine structures.

MgLa<sub>2</sub>S<sub>4</sub> (MgLa<sub>2</sub>Se<sub>4</sub>) as  $r_{\text{Ln}}$  increases in these spinels. The unfavourably expanded MgX<sub>6</sub> octahedra in the MgLn<sub>2</sub>X<sub>4</sub> spinels accounts for the destabilization of oct relative to tet as  $r_{\text{Ln}}$  increases.

One of the motivations for studying spinel conductors is that they flatten the migration energy profile by stabilizing tetrahedral Mg and enabling Mg to pass through a more stable octahedral configuration during migration. This principle suggests that minimizing the difference in energy between the oct and tet states would lead to the lowest possible migration barriers because the energy profile is maximally flat when  $E_{\text{oct}} = E_{\text{tet}}$ . However, within this group of MgLn<sub>2</sub>X<sub>4</sub> compounds, we find instead that maximizing this energy difference results in the lowest migration barriers (with the exception of MgLa<sub>2</sub>S<sub>4</sub>) because the area of the face-sharing triangle is largest (most stable) when the MgX<sub>6</sub> octahedra is largest (least stable).

Finally, the stability of the spinel structure relative to competing polymorphs and competing compounds was investigated using the revised-TPSS meta-GGA density functional.<sup>21</sup> Meta-GGA's have been shown to be superior to GGA in determining the ground-state structure<sup>24</sup> and stability<sup>25</sup> of chalcogenides. In Fig. 5, the energy above the convex hull of stable compounds in each chemical space,  $E_{\text{hull}}$ ,<sup>14,26</sup> is shown for all compounds in five different structures typically observed for AB<sub>2</sub>X<sub>4</sub> compounds.<sup>27</sup> Experimentally known structures for each compound are highlighted along the top of Fig. 5. Stable compounds have  $E_{\text{hull}} = 0$ , and synthesized sulphides and selenides have been observed to be metastable up to  $E_{\text{hull}} \approx 50$  meV per atom.<sup>28</sup> An even wider range of accessible metastability is expected for spinels because of their especially low surface energy, which may result in preferential nucleation during synthesis.<sup>29</sup>

MgLu<sub>2</sub>S<sub>4</sub>, MgLu<sub>2</sub>Se<sub>4</sub>, MgTm<sub>2</sub>Se<sub>4</sub>, and MgEr<sub>2</sub>Se<sub>4</sub> were calculated to have  $E_{\text{hull}} < 25$  meV per atom and spinel as the lowest energy structure at that composition. Additionally, MgTm<sub>2</sub>S<sub>4</sub>, MgEr<sub>2</sub>S<sub>4</sub>, and MgHo<sub>2</sub>Se<sub>4</sub> have  $E_{\text{hull}} < 50$  meV per atom in the spinel structure but MnY<sub>2</sub>S<sub>4</sub> is the lowest energy polymorph. On this basis, these seven materials are considered potentially synthesizable in the spinel structure. Indeed, all but MgEr<sub>2</sub>S<sub>4</sub> have been reported to crystallize in the spinel structure (MgTm<sub>2</sub>S<sub>4</sub> has been reported in both spinel

and MnY<sub>2</sub>S<sub>4</sub> structures).<sup>27,30</sup> Our calculations are generally consistent with the experimental observations as the experimentally observed crystal structure is calculated to be the lowest energy structure at MgLn<sub>2</sub>X<sub>4</sub> for 9 of 10 materials (all except MgHo<sub>2</sub>Se<sub>4</sub>), and all 10 experimentally observed structures are calculated to be stable or nearly stable ( $E_{\text{hull}} < 50$  meV per atom).<sup>27,30</sup>

The calculated sequence of stable crystal structures can be understood by considering the Ln coordination environment. At low  $r_{\text{Ln}}$ , spinel is the lowest energy structure, then MnY<sub>2</sub>S<sub>4</sub> becomes the lowest energy for intermediate  $r_{\text{Ln}}$ , and the Th<sub>3</sub>P<sub>4</sub> structure is the most stable polymorph for the largest Ln's. The Ln coordination number increases from 6 in spinel to 6/7 in MnY<sub>2</sub>S<sub>4</sub> to 8 in Th<sub>3</sub>P<sub>4</sub>, stabilizing the larger and larger Ln. The larger selenium stabilizes spinel for larger Ln, up to at least Ho, whereas spinel is the calculated ground-state only for Ln=Lu among the sulphides.

Increasing the size of the cation in chalcogenide spinels decreases the Mg<sup>2+</sup> migration barrier. However, increasing the cation size also leads to a preference for non-spinel structures and a general destabilization relative to competing compounds. The spinel structure has the advantage of placing Mg in its less stable tetrahedral configuration, decreasing the Mg migration barrier; however, this advantage is balanced by energetically preferred alternative structures for large cations. Still, we predict MgLu<sub>2</sub>S<sub>4</sub>, MgLu<sub>2</sub>Se<sub>4</sub>, MgTm<sub>2</sub>S<sub>4</sub>, MgTm<sub>2</sub>Se<sub>4</sub>, MgEr<sub>2</sub>S<sub>4</sub>, MgEr<sub>2</sub>Se<sub>4</sub>, and MgHo<sub>2</sub>Se<sub>4</sub> to be candidate solid-state electrolytes that strike a balance between relatively small Mg migration energies ( $E_{\text{mig}} < 380$  meV) and relatively good stability ( $E_{\text{hull}} < 50$  meV per atom) in the spinel structure.

The stability and Mg mobility of the investigated materials were calculated with density functional theory (DFT)<sup>31</sup> using the Vienna *Ab Initio* Simulation Package (VASP).<sup>32</sup> For mobility calculations, the generalized gradient approximation (GGA) according to Perdew, Burke, and Ernzerhof (PBE)<sup>33</sup> and the projector augmented-wave (PAW) method<sup>34</sup> were used with an energy cut-off of 520 eV for plane waves.  $2 \times 2 \times 2$  supercells, e.g., Mg<sub>16</sub>La<sub>32</sub>Se<sub>64</sub> and (Mg<sub>15</sub>La<sub>32</sub>Se<sub>64</sub>)<sup>2-</sup>, were calculated using a  $2 \times 2 \times 2$  Monkhorst-Pack  $k$ -point mesh. The positions of the atoms and the cell volume were optimized for the pristine structure. For the subsequent mobility calculations with one Mg<sup>2+</sup> vacancy, only the positions of the ions were optimized. Electronic and ionic relaxations were performed with convergence parameters of  $10^{-5}$  eV and  $10^{-2}$  eV Å<sup>-1</sup>, respectively. The Mg<sup>2+</sup> vacancy was charge-balanced by a neutralizing background charge. The minimum energy pathways were calculated by linearly interpolating seven images between the optimized initial and final structures and subsequently applying the nudged elastic band (NEB) method.<sup>35</sup> For the stability calculations, the revised-TPSS self-consistent meta-GGA<sup>36</sup> within the PAW method was used to calculate all MgLn<sub>2</sub>X<sub>4</sub> compounds in each of the five prototype structures – spinel, MnY<sub>2</sub>S<sub>4</sub>, Th<sub>3</sub>P<sub>4</sub>, olivine, and Yb<sub>3</sub>S<sub>4</sub> – and all relevant competing phases available in the Materials Project.<sup>37</sup> All calculated results are available as a table in the ESI.†

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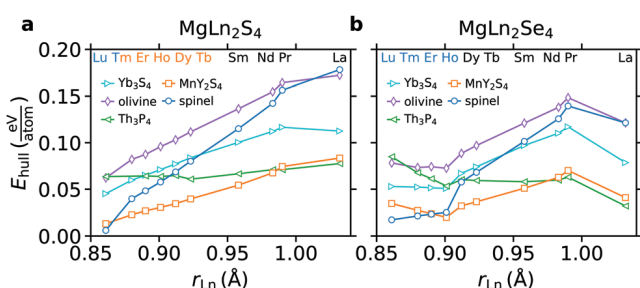


Fig. 5 Stability with respect to competing phases,  $E_{\text{hull}}$ , for each sulphide (a) and selenide (b) in a variety of crystal structures common to AB<sub>2</sub>X<sub>4</sub>-type compounds. The colour of each element at the top of both panels correspond with the experimentally observed crystal structure for that composition.<sup>27,30</sup> MgTm<sub>2</sub>S<sub>4</sub> is coloured for both spinel and MnY<sub>2</sub>S<sub>4</sub> because both crystal structures have been reported. Elements coloured black indicate no experimentally known crystal structure.

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## Conflicts of interest

There are no conflicts to declare.

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